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SEPARATION OF POLONIUM FROM BISMUTH BY DISTILLATION

By

R. W. Endebrock

P. M. Engle



August 1, 1953

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THE SEPARATION OF POLONIUM FROM BISMUTH BY DISTILLATION

(Information Report)

BY

R. W. ENDEBROCK AND P. M. ENGLE

Date: August 1, 1953

Work performed under Contract No. AT-33-1-Gen-53.

MOUND LABORATORY
Operated by
Monsanto Chemical Company
Miamisburg, Ohio

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ABSTRACT

The experimental work on the separation of polonium from bismuth by distillation during the period from 1949 to 1952 is reviewed. Data are presented showing operation of simple batch stills and of fractionating stills with and without helium sparging. The problem of materials of construction is discussed and data obtained while working on this phase of the project are included. Some of the problems of still design and construction are presented with recommendations for their solution.

INTRODUCTION

The investigation of separation of polonium from bismuth by distillation has progressed to the point where a prototype model of a still is under construction. At this point it seems appropriate to review the results of experimental work to date, much of which has been reported previously in abstract or progress-report form. The work reported in detail here is for the most part that of the Chemical Engineering Section and references are given to related work by other groups.

This work, which has been known as the Bismuth Distillation Project, has covered two general approaches to still operation; namely, simple distillation and fractional distillation. Both of these are batch processes and most of the work has been directed to fractionation. In addition, both this Laboratory and Battelle Memorial Institute have done considerable work on the corrosion of materials of still construction by heated bismuth. The details of still design necessary to obtain efficient, dependable operation have been studied carefully.

RESULTS OF EARLY DISTILLATION EXPERIMENTS

The original pilot experiments which were performed in glassware have been adequately reported. The best results were: depletion of 16 grams of irridated bismuth in the still-bottom of 98.3 per cent of its polonium in 28 hours, and 99.47 per cent in 42 hours; with polonium product concentrations of 35 and 25 times that of the original charge.

FRACTIONATING STILLS

A rather large Incomel fractionating still similar to that shown in Figure 1 was constructed. Before experiments with active bismuth were started in this unit, it was found that the Incomel was attacked by bismuth to such an extent that the unit was not usable, 2 This experience emphasized the need for better information on the corrosive effects of bismuth.

A still of the same design was then constructed of SAE 1020 steel and operated with active bismuth. One complete run of four cycles totalling 15½ hours was made and the data for this run are tabulated in Table I. During the second run with this equipment the still bottom failed by distortion from atmospheric pressure on the relatively-thin low-carbon steel still walls. This material has a very low creep-strength at the temperature employed, 825°C. The data indicate a high concentration factor but a rather low rate of product recovery. These results are probably due to the long narrow shape of the still column compared to that of the still bottom.

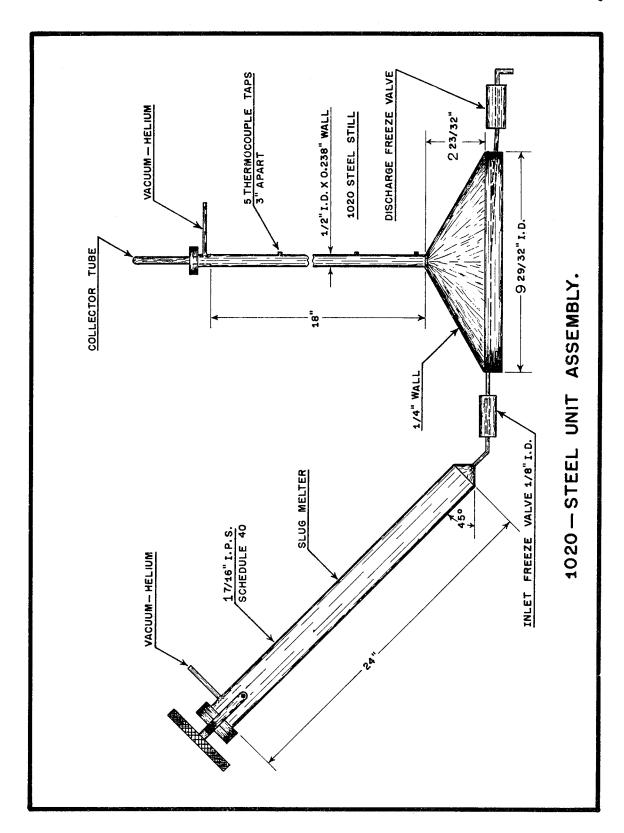


FIGURE 1

TABLE I
SUMMARY OF DATA, SAE 10:20 STEEL FRACTIONATING STILL

INITIAL CHARGE: 2,488 g Bi; 11.76 C Po

CYCLE	RUN TIME	PROD	ист	RE	ECOVERY	CONCENT	RATION FACTOR
NUMBER		Bi	Po	CYCLE	CUMULATIVE	CYCLE	CUMULATIVE
	(hr)	(g)	(C)	(%)	(%)		
1	3.5	0.212	3.63	30.9	30.9	3625	3625
2	5.0	0.288	4.64	39.4	70.3	4920	3510
3	4.5	0.535	1.27	10.8	81.1	1692	1950
4	2.5	0.380	0.19	1.6	82.7	561	1455

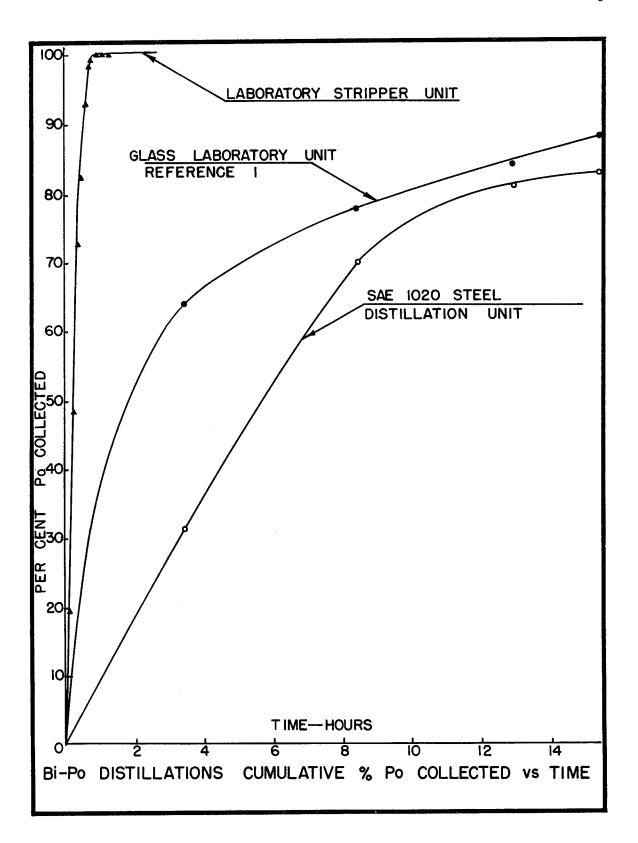
NOTES: STILL BOTTOM OPERATING TEMPERATURE, 825°C. OPERATING PRESSURE, 1 - 8 MICRONS

STRIPPING STILL

In order to demonstrate that irradiated bismuth could be depleted of its polonium content at a rapid rate by distillation, the stripping apparatus shown in Figure 2 was constructed. The irradiated bismuth charge was placed in the annulus formed between the steel outer wall and the central tube. The steel assembly was capped loosely and placed in the quartz housing. The housing was evacuated and the upper four inches of the steel stripper heated to about 1000°C by induction heat. Vaporized bismuth and polonium passed upwardly through the annulus and downwardly through the central tube to be collected in a small collector tube (not shown in Figure 2) in the cooled bottom of the quartz housing. After each heating cycle the collector was removed and the product was analyzed for polonium and bismuth. The results of a run consisting of 11 cycles are given in Table II. It was demonstrated that under sufficiently severe conditions bismuth can be depleted of its polonium content rapidly by distillation. The high temperature and absence of any fractionation or partial condensation effect resulted in large amounts of bismuth being present in the products and low concentration factors were obtained.

The rates of depletion for the various early experiments are shown in Figure 3, and Figure 4 shows the variation in concentration factor and degree of product collection obtained in these experiments.

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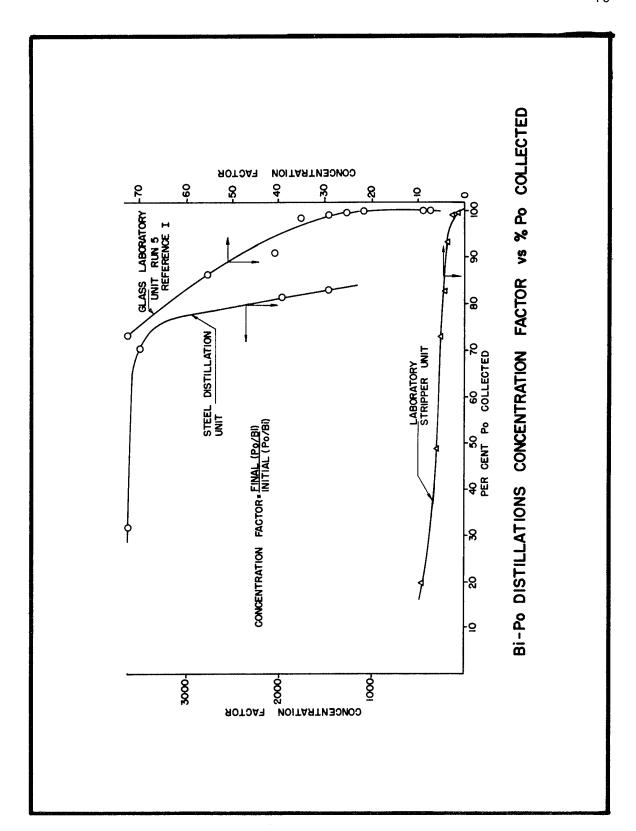


TABLE !!

SUMMARY OF DATA. STRIPPING STILL

INITIAL CHARGE: 215.8 g Bi; 2.001 C Po

CYCLE		MAX	COLLECTED	CUMULATIVE	COLLECTED	CUMUL AT I VE	CONCENTR	ATION FACTOR
NUMBER	RUN TIME	TEMP	Po	Po	Bi	Bi	CYCLE	CUMULAT: VE
	(Min)	(°C)	(C)	(%)	(g)	(%)		
1	6	850	0.3900	19.49	5.0376	2.33	8.33	8.33
2	8	1,025	0.5812	48.54	12.1347	8.15	6.27	6.10
3	8	986	0.4792	72.49	13.3108	14.3	6.94	5.13
4	8	1,025	0.1961	82.30	11.2821	19.6	5.86	4.26
5	9	1,070	0.2089	92.74	15.3197	26.6	6.69	3.51
6	7	1,072	0.1162	98.55	18.6803	35.4	6.76	2.81
7	4	>1,000	0.0224	99.67	21.0562	45.1	4.74*	2.22
8	7	>1,000	0.00485	99.91	22.4086	55.5	3.39*	1.81
9	8	>1,000	0.00154	99.98	20.1744	64.8	2.69*	1.55
10	7	>1,000	0.000177	99.998	22.4900	75.4	- *	1.33
11	6	>1,000	0.0000261	99.999	13.4590	81.5	- *	1.23

NOTES: PRESSURE WAS APPROXIMATELY 1 MICRON AT START OF EACH CYCLE.

$$\text{CONCENTRATION FACTOR} = \frac{\frac{\text{g Bi}}{\text{C Po}}}{\frac{\text{g Bi}}{\text{C OLLECTED}}} \frac{\text{g Bi}}{\text{C Po}}$$

AGITATION

Various methods were considered for improving still performance obtained in the early experiments. One of the most promising methods appeared to be that of agitation of the bismuth pool in the still bottom. Since the rate of polonium volatilization from the surface of the bismuth should be very rapid at the temperatures employed, it was assumed that the rate-controlling mechanism was the rate of diffusion of the polonium from the body of the melt to the surface. Experimental work described below under the heading Simple Distillation Experimental Work confirmed this assumption, at least to a degree. Several methods of ordinary mechanical agitation were investigated briefly, but these methods were not considered to be practical because of high still-temperature and vacuum requirements.

^{*}VALUES ARE NOT ACCURATE BECAUSE OF LOSS OF SIGNIFICANT FIGURES IN CALCULATION.

Some of the early work on distillation of polonium from irradiated bismuth indicated that small quantities of a gas or gases were liberated during distillation. Verification of this observation was made and the gases were studied at this Laboratory and at Battelle Memorial Institute. 4 The effect of liberation of gas during distillation was studied and it appeared to be beneficial. It was eventually incorporated in the design of distillation units having sparging inlets.

AGITATION TEST UNITS

Two test-units were constructed for the purpose of investigating the effect of agitation of the bismuth upon distillation. 5 One of these stills was a cylindrical unit divided into four compartments, each compartment of which was equipped with a separate identical product collection tube. After the first compartment was loaded it acted as a control means so that distillation occurred without agitation. By manipulating the pressures in each compartment, the bismuth was sprayed completely into each successive compartment. The distillation rate, expressed in millicuries of polonium per minute, was compared for each compartment. Depletion was not an important factor since the overall percentage of polonium distilled was small. The results are shown in Table III.

TABLE HILL
DISTILLATION RATE

COMPARTMENT	RATE (<i>mC/min</i>)
1 (NO SPRAYIN	, , ,
2 (SPRAY)	17.60
3 (SPRAY)	2,90
4 (SPRAY)	12.00

This test showed a maximum distillation rate 24 times greater than the rate without agitation, and an average distillation rate 15 times greater than that without agitation. The increased surface caused by the spraying would, however, account for a portion of this increased rate.

The other still was constructed from a Pyrex cylinder having a fritted Pyrex disk sealed in the bottom. Helium was passed through the disk and into the molten bismuth causing gas agitation or sparging. A pressure of approximately 10 microns above the bismuth permitted sufficient gas to flow into the bismuth so as to give a gentle agitation. The bismuth was heated by an induction heater which caused the metal to be heated to a higher temperature than the glass still. The method of product collection proved to be subject to a variety of errors. However, based on bismuth depletion rather than product recovery, the data shown in Table IV were obtained.

TABLE IV

RESULTS OF TEST OF STILL

Po in charge - 1.480 C

DEPLETION - 71.8%

PO REMAINING - 0.418 C

DIST. RATE - 17 MC/MIN

DIST. TIME - 60 MIN

Here again a rather high distillation rate was obtained. In another test employing a fritted Pyrex disk, it was found that the rate of polonium recovery when helium was being sparged was four times that without sparging. In these tests, however, the agitation was exceedingly gentle as compared to the violent agitation and increased surface area caused by spraying. The optimum flow of helium needed to maintain a maximum distillation rate has not been established. At very low sparging rates the surface of the molten bismuth is not preceptibly broken, consequently there is doubt concerning the mechanism involved in the introduction of helium to the bottom of the bismuth melt. The mechanism may be a diffusional process rather than a true agitation or may be similar in nature to steam distillation. The results of later non-sparging runs with the sparging still cast some doubt on the efficiency of sparging. Further investigation of the effect is warranted.

EXPERIMENTAL SPARGING STILLS

Two fractionation units with inlets for helium sparging were constructed as shown in Figure 5, one of low-carbon steel and the other of 3 per cent silicon transformer steel. Both units were operated successfully for several hundred hours and considerable design data and operating technique were obtained.

OPERATING PROCEDURE

An irradiated slug of bismuth was calorimetered and used as a still charge. The weighed charge was loaded into the still from a melter and a small quantity of non-irradiated bismuth was used to seal the loading arm. The unit was then cooled and the condenser was inspected for spattering. If spattered metal was evident it was removed and assayed for bismuth and polonium. The still was brought to temperature in approximately one hour. An initial pressure reading was taken before sparging was begun and an initial pressure of less than one micron during heat-up indicated that the vacuum system was tight. This pressure reading served as a reference value for control of sparging rate. The sparging rate was adjusted to allow an increase of 2 to 5 microns in the pressure reading, and this rate was maintained throughout the run and during the cool-down period. When a temperature of 300°C had been reached by the still pot sparging was discontinued. Pressure readings were taken until the still temperature was below 200°C.

After the still had cooled the product condenser was removed and inspected for signs of oxidation which was indicated by discoloration of the condenser. An inspection for the presence of polonium on the outside surface of the condenser was made with zinc sulfide paper to make sure that complete collection had occurred on the inside surface. No appreciable outside contamination was found during the runs. The product was then carefully dissolved in concentrated nitric acid, placed in a new 100-milliliter volumetric flask and adjusted to a

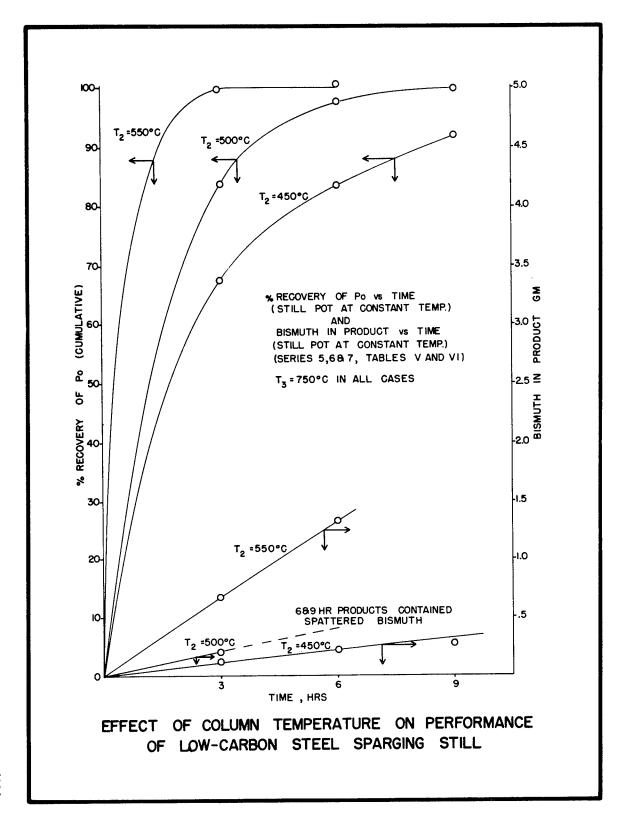
concentration of approximately four-normal with nitric acid. A sample was sent to the Micro-Assay Group for determination of bismuth, and a second sample was mounted with special precaution being exercised to maintain clean micro-pipettes. This mount was counted. Every effort was made to keep counting errors at a minimum, and known samples were frequently counted as control blanks. As a result of these precautions, material balances were generally well within counting error. The still was unloaded, the spent bismuth was weighed, and a small sample of the bismuth weighing one to three grams was dissolved in 100 milliliters of four-normal nitric acid. A sample of this solution was mounted and counted. Again precautions were exercised to obtain reliable results. An assay of the spent bismuth provided a method for cross-checking the recovery, and was subject to less error than an assay of the product solution. Large percentage errors in counting low-activity solutions gave less overall error in material balances than small errors in counting of high-activity product solutions. A weighed sample of the dissolved spent bismuth was sent to Micro-Assay Group as a control-check on the thiourea colorimetric method used for the determination of bismuth in the product.

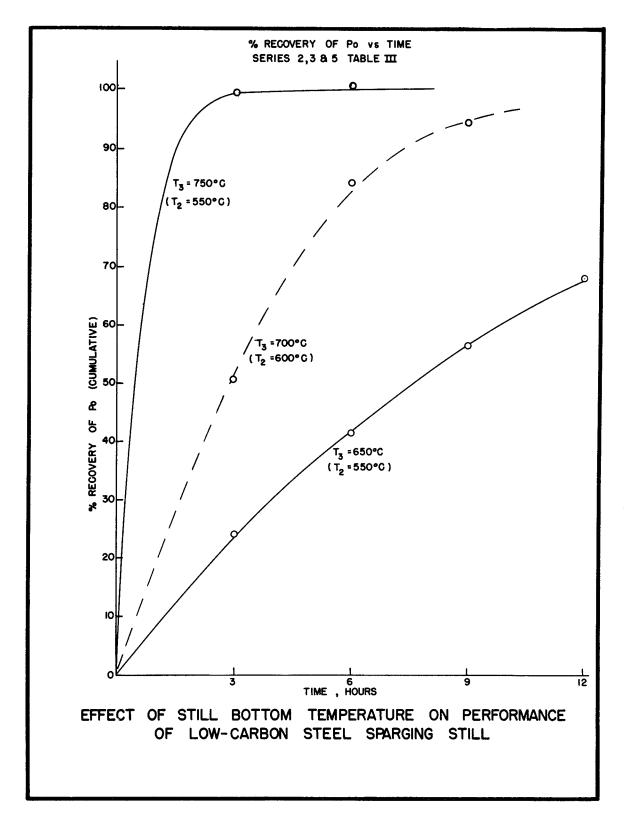
LOW-CARBON STEEL SPARGING STILL

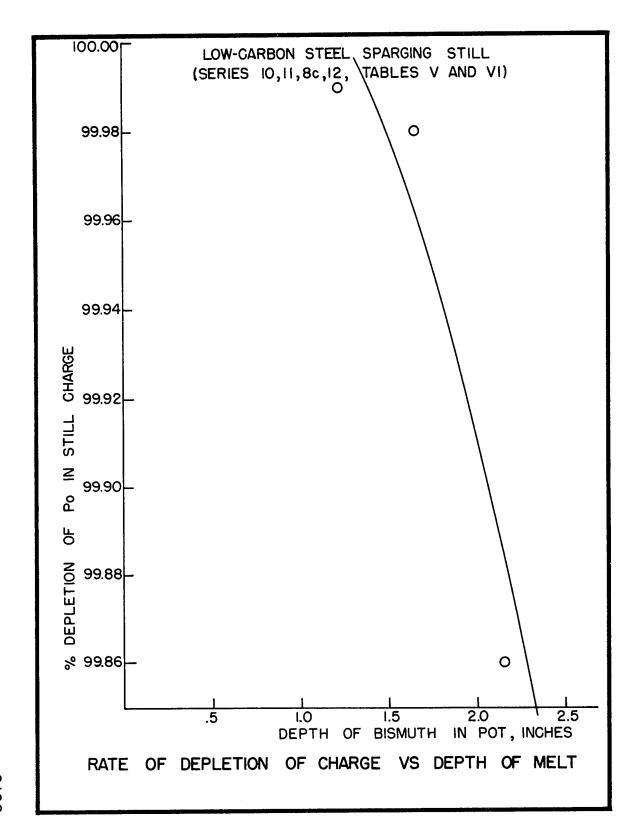
Data obtained from the low-carbon steel unit included: (1), optimum operating temperatures of the column and still-bottom for a three-hour cycle; (2), the general effects of variations of column and still-bottom temperatures on distillation; (3), partial data on the effect of column constrictions; and (4), partial data on the effect of variations in bismuth-depth on performance.

Tables V and VI list the operating data for this still. Temperatures were taken at the top of the column, T1; at the middle of the column, T2; at the middle of the still-pot, T3; and at the bottom of the still-pot, T4. For ease of operation, the middle of the column and the middle of the still pot defined the control conditions of temperature. Figure 6 shows the relationship between column temperature and the amount of both polonium and bismuth collected, with temperature, of the still bottom held constant at optimum operating temperature. As would be expected the polonium curves approximate the form of an exponential curve while the bismuth curves are straight lines. Figure 7 shows the relationship between still-pot temperature and product recovery with the column temperature held approximately constant at optimum temperature. Within limits it was found that the column temperature controls product purity while the still-pot temperature controls distillation rate. Figure 8 is a graph of the limited data on the effect of bismuth depth on distillation rate expressed as per cent depletion of the charge. Because of the narrow range of depletions and depths covered by the data, this plot is considered to be a qualitative measure of the variation in depletion rate with changes in bismuth depth. The effect of depth might be better expressed in terms of the time required to reach a given degree of depletion, operating conditions being held constant at a still bottom temperature somewhat lower than optimum. Equipment now on hand should permit a more complete study of the depth effect.

Figure 9 shows approximately the relationship between product concentration and column temperature with the still-pot temperature held constant at 750°C and a distillation time of three hours for each point. A better illustration of the relationship would have been to plot product concentration versus column temperature for the same product recovery with a constant still-pot temperature. Extrapolation of the data as shown by the dotted curve of Figure 9 indicates that for these conditions considerable time would be sacrificed for a minor gain in product concentration by operating the column below 550°C.







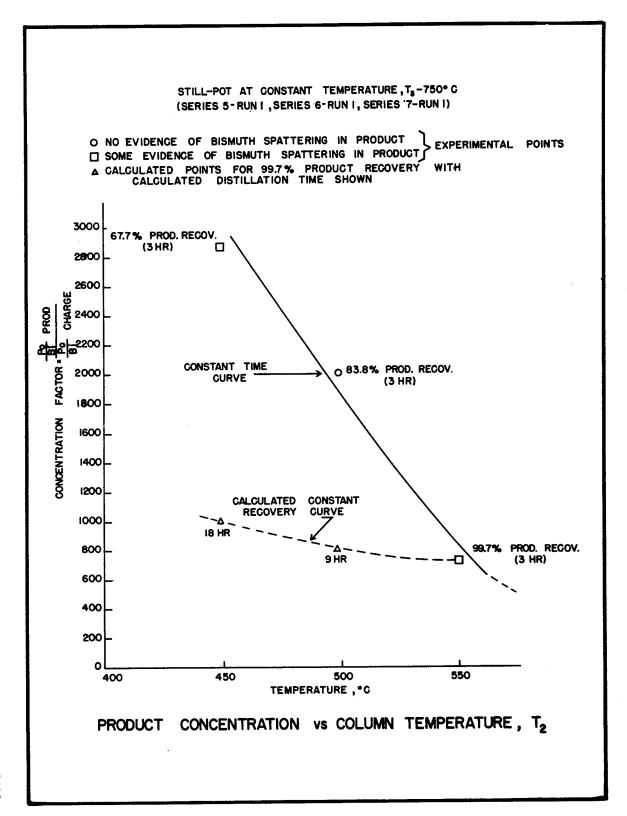


TABLE V
OBSFRVED OPERATING DATA FROM LOW-CARBON STEEL SPARGING STILL

	Po IN DISCHARGED BI (C)	1 1 1 1	0.982	1. 514 X 10 ⁻³	1 1 1	SAMP, LOST	3.39 × 10 ⁻⁴	1 1 1	2.1 × 10 ⁻³	1.65 × 10 ⁻³	9.66 X 10 ⁻³	2.27 X 10 ⁻⁴	1.17 × 10 ⁻³	1.22 x 10 ⁻²	3.702 X 10 ⁻²	3.60 × 10 ⁻²	
	PRODUCT Po (C)	0.098 0.302	0.987 0.711 0.617 0.480	3.38 2.26 0.69 0.244 0.019	1.62 1.25 0.952	2.065	3.835 0.903 0.478 0.450	4.180 0.690 0.105	7.145	3.56	6.737	2.704	4.65	3.158 0.766 0.157	3,45	2.781	
SPARGING STILI	PR Β1 (g)	0.78 0.70 0.68 1.10	1.4 0.95 1.26 2.27	2.9 3.25 3.25	5.59 3.79 2.60	0.67	0.12 0.09 0.05	0.20 1.5 1.0	1.8	1.4	1.5	1.5	1.16	2.0 0.9 0.68	1 . 0	1.4	
	CHARGE Po (C)	4.499 4.401	4.098 3.080 2.369 1.734	6.678 3.215 0.954 0.264	5.993 4.373 3.122	2.071	5.667 1.832 0.929 0.444	4.990 0.810 0.121	7.308	3.647	6.917	3.267	4.808	3.919 0.761 ?	3.220	2.929	
RBON STE	CH/ Bi (g)	400 400 463.5 462.8	461.7 460.2 459.2 457.9	532.0 529.1 525.0 521.1 515.8	493.4 487 8 484.0	481.4	507.1 507.0 506.9 506.4	478.9 478.7 477.2	672 670.2	381.5	687.8	380.9	520.2	533.9 531.9 531.0	520.7	503.6	
DATA FROM LOW-CARBON STEEL	VACUUM VI PHILLIPS (μ)	14 2.3.4 35 4.5 40 5.8 35 4.8	12 1-3 12 1-4 10 0, 5-4 0.5	5-7 0.5-2 5-7 0.5-3 2-4 0.5-1 5-12 0.4-4 2-13 0.2 4	0.7 0.69 0.6-1	0.8-2	0.0 0.8 2 2 2 2	ଅଧନ	3.5	-	1.5	1.6	ю	3.5 3 20	0.35	0.4	
TA FF	ν, PIRANΙ (μ)	5.14 16.35 10.40 25.35	3.12 3.12 2.10 0.5	0.0	000 00	4 m	пппп	14 4.5	5.5	2.5	3,5	က	2.3	4.5 4 20	2.7	4.7 REASONS	
.ING	$\mathbf{T_4}$ POT BOTTOM (C°)	545 545 545 540	590 590 590 590	630 630 630 630 630	590 590 590	675 680	670 675 675 675	675 675 675	675 680	680	675	675	675	670 670 680	670	670 VAR10US	
OP ER/	$\begin{array}{c} \mathbf{T_3} \\ \text{POT} \\ \text{CENJER} \\ \langle \mathcal{O} \rangle \end{array}$	600 600 600 600	650 650 650 650	700 700 700 700 700	6 50 6 50 6 50	7 50 7 50	7 50 7 50 7 50 7 50	750 750 750	7 50 7 50	7 50	7 50	7.50	7 50	750 750 750	740	750 LETED FOR	
BSERVE	$\frac{\mathbf{T}_{2}}{COLUMN}$ CENTER (\mathcal{C}^{2})	500 500 500	550 550 550 550	009	009 009	550 550	450 450 450 550	500 500 500	550 550	550	550	550	550	550 550 550	550	90 540 750 670 WERE NOT COMPLETED FOR VARIOUS	
0	$\begin{array}{c} \mathbf{T_1} \\ \mathrm{COLUMN} \\ \mathrm{TOP} \\ (C^{\circ}) \end{array}$	370 370 370 370	400 400 400	425 425 425 425 425	430 430 430	390 390	325 330 325 395	360 360 360	395 395	395	395	395	395	400 400 395	390	. 8 8	2
	RUN TIME (HTS)	ოოოო	ოოოო	ოოოოო	е е е	ოო	пппп	ოოო	ကက	m	m	ю	က	ოოო	6	3 AND	
	SERIES & RUN (No)	TEST TEST 1.1 1.2	2.2 2.3 2.4		4-4 4-2 8-3	5-2	6.1 6.2 6.3 CLEAN UP	7.17.27.3	8.1	9-1	10.1	1-11	12-1	15-1 15-2 15-3	16.1	17.1 SERIES 13.	,

TABLE VI

CALCULATED OPERATING DATA LOW-CARBON STEEL SPARGING STILL

REMARKS	NON-SPARGING COLD RUN SPARGING COLD RUN STILL LOADED WITH ACTIVE BISMUTH CHARGE CARRIED OVER FROM SERIES 1 STILL DISCHARGED	STILL LOADED STILL DISCHARGED	STILL LOADED CHARGE CARRIED OVER FROM SERIES 4. EVIDENCE OF	SPATIERING STILL DISCHARGED STILL DOGED. EVIDENCE OF SPATTERING NO EVIDENCE OF SPATTERING STILL DISCHARGED	STILL LOADED. NO EVIDENCE OF SPATTERING VAC LEAK, PRODUCT SPATTERED STILL DISCHARGED, SPATTERING - POSSIBLE VAC LEAK 2.15" B1 DEPTH. B1 IN ANNULUS F0 IN DISCHARGED B1 = 0.0005%, DRICH ANNULUS. STILL	LOADED AND DISCHARGED ON EACH SUBSEQUENT KUN. 1.21" DEPTH, VAC LEAK - SPATTERING, BI IN ANNULUS 2.15" BI DEPTH, STILL DISCHARGED. BI IN ANNULUS 1.21" BI DEPTH, SPATTERING PRODUCT PARTLY SPILLED.	BI IN ANNULUS VAC GAUGE ERROR - VAC LEAK, BI IN ANNULUS COLUMN BAFFLE INSERTED AND REMAINED UNTIL THIS STILL WAS INOPERATIVE. SOME BI IN ANNULUS. PRODUCT PARTLY	SOME BI ANNULUS VAC LEAK. SOME BI IN ANNULUS	NO SPARGING VAC LEAK. NO SPARGING	
CONCENTRATION FACTOR CYCLE CUMMULATIVE	- 23 72 72 54	932 277	22 255 7 66 7	365 2860 2019 1795 0VERALL	2005 2006 175 365 198	266 447 210	434 215	184 155	558 342	REASONS
CONCENT	7 200 112 129 129 130 130	93 90 97 92 153	24 37 716	338 2860 2777 522 9000	2006 272 414 365 414	266 447 210	434 215	595	558 342	VARIOUS
RECOVERY CUMULATIVE (%)	2.19 2.19 24.09 56.44 68.00	50.61 84.46 94.79 98.44	27.03 47.89 48.09 99.71	101.04 67.67 83.61 92.04 overall	993.77 997.77 997.77 77	97.62 97.40 82.77	96.71 80.58	100,13	107,14 94,95	NOT COMPLETED FOR
REC CYCLE (%)	24.09 23.08 26.05 27.68	50.61 70.30 72.33 92.42 95.00	27.03 28.59 30.49 99.71	458.33 67.67 49.29 51.45	83.77 85.19 86.78 97.77 98.77	97.62 97.40 82.77	96.71 80.58	100,66	107.14 94.95	WERE NOT
DEPLETION (%)		78, 17 - - - - - 0VERALL	/ 6 · 66 6	- - OVERALL	99,99+ - - - 0VERALL	99.97 99.95 99.86 99.99	99.98	OVERALL	99.16 98.77	14 AND 18
SERIES AND RUN (#0)	TEST TEST 1-1- 2-2- 2-3	ఴ ఴఴఴఴ ⊷ઌఴౚఴ	4440 -96-	5-2 6-1 6-3 CLEAN UP	7.77 7.22 8.33 8.2	9-1-0-1 1-1-1	15-1	15-2 15-3	16-1	SERIES 13,

A baffle was inserted in the bottom of the still column in an attempt to eliminate bismuth spattering during still operation. Although the baffle constricted the column to less than one-eighth of its cross-sectional area distillation rates and yields appeared to be unaffected. The still failed before a complete study could be made of the effect of reduced column cross-sectional area.

The series marked 16 and 17 in Tables V and VI were run without sparging. Although the pressure readings indicate that a reasonably good vacuum was obtained there is a possibility that a small leak occurred through the sparging arm. These data indicate that the lack of sparging had little effect upon rate of distillation and create some doubt as to reliability of earlier sparging tests. A positive comparison of sparging and non-sparging will be made with new equipment now on hand.

Examination of Tables V and VI indicate erratic concentration values. These variations probably were caused either by bismuth being spattered to the condenser or by adhering to the condenser of some previously-spattered bismuth in the annulus. Improved still design should alleviate if not eliminate this condition

SILICON STEEL SPARGING STILL

Battelle Memorial Institute tested a large number of materials for use in the construction of bismuth stills. The most promising of the alloys tested was 3 per cent silicon steel, commonly called transformer steel Battelle reported that silicon steel was not wetted by bismuth in the range of anticipated still temperatures, and that the corrosion rate caused by bismuth was exceedingly low A distillation unit was constructed of this alloy from the design drawings of the low carbon steel unit in order to evaluate this material under actual operating conditions.

Comparing the data shown in Tables VII and VIII for the silicon steel unit with that shown in Tables V and VI for the low-carbon steel still, a lesser rate of heat transfer from the still walls to the bismuth is indicated for the silicon steel unit. Although higher product concentrations were obtained, lower product recoveries were found. However, still-bottom depletions were about the same as for the low-carbon steel still. In order to approach the results obtained with the low-carbon steel unit, higher still wall temperatures were required. Since the creep-strength for transformer steel and low-carbon steel are in the same range, design considerations would require a thicker still wall if transformer steel alone were used for the body of a large still. Of greater concern were the low and erratic product yields which indicated possible interaction between polonium and transformer steel.

THE STILL WAS LOADED AND DESCHARGED FOR EACH RUN

TABLE VII

OBSERVED OPERATING DATA SILICON STEEL SPARGING STILL

	Po in	DISCHARGED B1	(0)	3.89 x 10 ⁻²	5.686 x 10 ⁻²	5,40 x 10°8	2 892 x 10 ⁻³	3 033 x 10 ⁻²	775 × 103	1.31 × 10 ⁻²	2 69 × 10 ⁻²	0.149	3.87 x 10 ⁻²	6.415 x 10 ⁻³	2,439 x 10 ⁻²	3.729 x 10 ⁻²	2.643 x 10 ⁻²
	UCT	Po	(0)	2.15	7.94	8.729	8.234	7.754	7,258	7,558	6.822	690'2	9.921	10.02	9.028	8.37	10, 114
	PRODUCT	Bi	(8)	0.17	0.49	0.75	0.50	0.885	0.665	1.03	2 8	2.1	3.6	6.1	0.75	2.00	1.00
	3GE	Po	(0)	2.334	8.930	8.919	8 620	8,495	7 974	7,743	7,463	7.604	10, 245	10.356	9,563	9,348	10,549
	CHARGE	Bi	(8)	472.0	503.4	516.1	507, 5	514.2	521.6	518.7	513.6	532.9	537.0	547.7	550,2	561.1	599.8
	VACUUM	PHILLIPS	(m)	3-25	4	4	4	ო	4	4	3 5	4	ო	ക പ	4	က	ო
	VAC	P (RAN)	(\mathcal{H})	4-25	Ξ	12	12	Ξ	13	10	8.5	8.6	6	10.5	12	9	9
	T.4 Pot	BOTTOM	(6)	069	680	710	700	705	695	705	735	735	7.40	740	735	740	740
TEMPERATURE	T ₃ Pot	CENTER	(00)	750	750	775	770	775	775	800	820	820	825	825	825	825	825
TEMPE	T ₂ COLUMN	CENTER	(00)	550	009	909	290	575	009	625	650	620	625	625	625	625	625
	T ₁ COLUMN	TOP	(00)	360	390	390	380	380	405	410	415	410	410	405	405	400	400
	RUN	TIME	(hrs)	က	ю	က	4	ဇ	ю	က	က	ო	ო	ហ	ю	ო	ო
	SEB	AND RUN	(No)	1-1	2-1	3.1	1.1	5.1	6. 1	7.1	8.1	9.1	10-1	:	12-1	13.1	1 4 . 1

TABLE VIII

	REMARKS										S	S	S	PILLED PROD SI STEEL TEST ROD	EST ROD
NG STILL		VAC LEAK									Bi IN ANNULU	Bi in ANNULU	Bi in ANNULU	VAC LEAK - SI	LOW CARBON TEST ROD
	CONCENTRATION FACTOR	2558	914	674	026	530	714	492	168	236	145	294	693	251	575
SIL	RECOVERY (%)	92.12	88,91	97.87	95.52	91.28	91,02	97,62	91,41	95.96	96,84	96.76	94.41	89.54	95.88
	DEPLETION (%)	98.33	98.36	99.93	96.66	99.64	06°66	99.83	69.63	98.04	99.62	99,94	99.75	09 '66	99.75
	SERIES AND RUN (No)	1-1	2.1	3.1	4-1	5-1	6.1	7-1	8-1	9.1	10-1	<u>-</u>	12-1	13.1	14.1
	SILICON STEEL SPARGING STILL	SILICON STEEL SPARGING STILL DEPLETION RECOVERY CONCENTRATION FACTOR (\$) (\$)	SILICON STEEL SPARGING STILL DEPLETION RECOVERY CONCENTRATION FACTOR (\$) (\$) (\$) 98.33 92.12 2558 VAC LEAK	DEPLETION RECOVERY CONCENTRATION FACTOR (%) (%) 98.33 92.12 2558 VAC LEAK 99.36 88.91 914	SILICON STEEL SPARGING STILL DEPLETION RECOVERY CONCENTRATION FACTOR (%) (%) 98.33 92.12 2558 VAC LEAK 99.36 88.91 914 99.93 97.87 674	DEPLETION RECOVERY CONCENTRATION FACTOR STILL 98.33 92.12 2558 VAC LEAK 99.36 88.91 914 674 99.96 95.52 970 970	DEPLETION RECOVERY CONCENTRATION FACTOR STILL 98.33 92.12 2558 VAC LEAK 99.36 88.91 914 ACCOUNTRATION FACTOR 99.95 95.52 970 ACCOUNTRATION FACTOR 99.64 95.52 970 ACCOUNTRATION FACTOR 99.64 91.28 530	DEPLETION RECOVERY CONCENTRATION FACTOR STILL 98.33 92.12 2558 VAC LEAK 99.36 88.91 914 ACCOUNTRATION ACCOUNTRATION 99.96 97.87 674 ACCOUNTRATION ACCOUNTRATION 99.96 95.52 970 ACCOUNTRATION ACCOUNTRATION 99.96 91.28 530 ACCOUNTRATION ACCOUNTRATION 99.90 91.02 ACCOUNTRATION ACCOUNTRATION ACCOUNTRATION	DEPLETION RECOVERY CONCENTRATION FACTOR STILL 98.33 92.12 2558 VAC LEAK 99.36 88.91 914 ACTOR 99.96 97.87 674 ACTOR 99.64 91.28 530 99.90 91.02 714 99.83 97.62 714	DEPLETION (%) RECOVERY (%) CONCENTRATION FACTOR STILL 98.33 92.12 2558 VAC LEAK 99.36 88.91 914 ACTOR 99.95 97.87 674 ACTOR 99.64 91.28 530 ACTOR 99.83 97.62 A42 ACTOR 99.63 91.41 ACTOR ACTOR	DEPLETION RECOVERY (%) CONCENTRATION FACTOR STILL 98.33 92.12 2558 VAC LEAK 99.36 88.91 91.4 ACTOR 99.96 95.52 97.0 ACTOR 99.64 91.28 530 ACTOR 99.93 91.02 71.4 ACTOR 99.63 91.41 ACTOR ACTOR 99.63 91.41 ACTOR ACTOR 99.63 91.41 ACTOR ACTOR 98.04 92.96 ACTOR ACTOR	DEPLETION RECOVERY CONCENTRATION FACTOR STILL 98.33 92.12 2558 VAC LEAK 99.36 88.91 91.4 ACTOR 99.96 95.52 97.8 ACTOR 99.96 91.28 ACTOR ACTOR 99.97 91.02 ACTOR ACTOR 99.83 97.62 ACTOR ACTOR 99.63 91.41 168 ACTOR 98.04 92.96 ACTOR ACTOR 99.63 96.84 ACTOR ACTOR 99.62 96.84 ACTOR ACTOR 99.62 96.84 ACTOR ACTOR	DEPLETION RECOVERY CONCENTRATION FACTOR ST.ILL 98.33 92.12 2558 VAC LEAK 99.36 88.91 91.4 Accountration Accountration 99.36 97.87 674 Accountration Accountration 99.96 91.28 714 Accountration 99.63 91.02 714 Accountration 99.63 91.41 168 Accountration 99.67 96.84 145 Bi in annulus 99.94 96.76 83 in annulus	DEPLETION (\$4) RECOVERY (\$4) CONCENTRATION FACTOR STILL CON STEEL SPARGING STILL 98.33 92.12 2558 VAC LEAK 99.36 88.91 91.4 ACCOUNTRATION FACTOR 99.36 97.87 67.4 ACCOUNTRATION FACTOR 99.36 97.87 67.4 ACCOUNTRATION FACTOR 99.63 91.28 97.6 ACCOUNTRATION FACTOR 99.64 91.28 ACCOUNTRATION FACTOR ACCOUNTRATION FACTOR 99.64 91.28 ACCOUNTRATION FACTOR ACCOUNTRATION FACTOR 99.67 91.28 ACCOUNTRATION FACTOR ACCOUNTRATION FACTOR 99.67 96.76 ACCOUNTRATION FACTOR BI IN ANNULUS	DEPLETION RECOVERY CONCENTRATION FACTOR STILL 98.33 92.12 CONCENTRATION FACTOR 99.36 92.12 CONCENTRATION FACTOR 99.95 92 92.12 CONCENTRATION FACTOR 99.95 92 92.12 CONCENTRATION FACTOR 99.64 92.12 CONCENTRATION FACTOR 99.65 92.12 CONCENTRATION FACTOR 99.65 92.12 CONCENTRATION FACTOR 99.65 92.12 CONCENTRATION FACTOR 99.66 92.12 CONCENTRATION FACTOR 99.66 92.12 CONCENTRATION FACTOR 99.67 92.12 CONCENTRATION FACTOR 99.68 92.12 CONCENTRATION FACTOR 99.69 92 92 92 92 92 92 92 92 92 92 92 92 92

POLONIUM ABSORPTION ON SILICON STEEL

In order to establish the effect of transformer steel on polonium, rods of transformer steel and SAE 1020 steel were used in qualitative penetration tests. Each rod was placed inside the silicon steel still so that one end was submerged in the melt and the other end extended midway into the column. The still charges were of the same magnitude, and the runs were made under the same operating conditions, series 13 and 14 of Tables VII and VIII.

After the runs were completed, an attempt was made to decontaminate the rods by identical procedures. Comparisons of contamination on the rods during the decontamination process were made with a counting-rate monitor. Counting was done over the most active 1.5 inches of each rod.

The first step consisted of scraping off the bismuth adhering to the rod, and then removing 0.001 inches of the base metal with coarse emery cloth followed by metal polish. The second step consisted of removing another 0.001 inch of base metal with emery cloth and metal polish. Steps three through six consisted of polishing with metal polish. The sixth step included a wash in dilute hydrochloric acid, dilute bleach, and soap solution. At the conclusion of the test both rods were heated by means of an induction heater to approximately 900°C under vacuum for 3 minutes. The results of the tests are shown in Table IX. The data indicate that roughly ten times more activity was absorbed in the silicon steel rod than in the SAE 1020 steel rod under like conditions.

TABLE IIX

COMPARATIVE DECONTAMINATION OF SILICON STEEL AND SAE 1020 STEEL

STEP (<i>No)</i>	3% SILICON TRANSFORMER STEEL (cts/min)	SAE 1020 STEEL (cts/min)	REMARKS
1	2.3 X 10 ⁸	2.25 × 10 ⁵	0.001 IN. METAL REMOVED
2	14,950	1,600	0.001 IN. METAL REMOVED
3	20,700	2,800	METAL POLISH
4	23,000	1,760	METAL POLISH
5	24,100	1,760	METAL POLISH AND WASH
6	28,750	2,400	METAL POLISH AND FLASH
AFTER HEATING	23	23	

SIMPLE DISTILLATION EXPERIMENTAL WORK

The fractionating or partial condensation stills described in the preceeding sections use columns to increase the concentration of the product over that obtainable by simple distillation. In general, the terms "simple equilibrium" and "differential distillation" have

been employed practically synonymously in reports on this work. Due to the possible simplification of equipment and probability that results could be more easily predicted from uninvolved tests, simple distillation was investigated as a possible means of separating polonium from irradiated bismuth.

After the fractionating work achieved the excellent results shown in Tables V and VI, the simple distillation work was discontinued although related fundamental research has continued to a certain extent.

RELATIVE VOLATILITIES AND DISTILLATION RATES

Fundamental experiments on the relative volatility α , also referred to as the enrichment factor K, of polonium from polonium-bismuth mixtures have been performed by R. W. Moshier and E. F. Joy of the Chemistry Section, 6,7,8,9,10 This work includes data on the vapor pressure of bismuth and its rate of evaporation. The best values available for these quantities, based on the preceding references and some unpublished data by Joy, are as follows:

Relative Volatility,
$$\alpha = -4.9828 + \frac{18219}{T + 1728}$$
 (unrestricted)

Relative Volatility,
$$\alpha = -3.0073 + \frac{10348}{T + 1150}$$

Data for computing the second equation was obtained with orifices between the distilling mixture and the condenser to reduce the rate of distillation. It should represent equilibrium conditions since the amount of restriction appeared to be immaterial when the orifices were varied from about 0.1 per cent to 5 per cent of the area of the distilling bismuth.

Distillation rate of bismuth
$$\log w = 8.3878 - \frac{10051}{T} - \frac{1}{2} \log T$$
 (low pressure)

Vapor pressure of pure bismuth,
$$\log P = -\frac{10051}{T} + 8.462$$

Vapor pressure of pure polonium,
$$\log P = -\frac{5378}{T} + 7.2345$$

The last equation is taken from the work of Brooks. 11 The temperature limits of all five of the above equations are from 450° to 850° C.

In the above equations,
$$\alpha = \frac{\log \frac{(\text{Po}) \text{ initial}}{(\text{Po}) \text{ final}}}{\log \frac{(\text{Bi}) \text{ initial}}{(\text{Bi}) \text{ final}}}$$

T = degrees Kelvin

w = grams per square centimeter per second

P = millimeters of mercury

SIMPLE DISTILLATION PROPOSALS

Based on some of these results, calculations were made to show the feasibility of the simple distillation process. As an example of the results of these calculations, a two-still simple distillation system could give a product 100 times more concentrated than the feed while recovering 99.5 per cent of the polonium.

The simple distillation of polonium from bismuth by flashing was considered 13 This approach did not appear to be promising unless very fine subdivision of the molten bismuth could be obtained. An information report on simple batch distillation was prepared at this time. 14 It suggests a falling-film type of molecular distillation as a method of separation which should be investigated.

EXPERIMENTAL RESULTS

Determinations of relative volatility and evaporation rate were made by the Chemical Engineering Section employing equipment from simple distillation applications. 4, 15 An example of the type of equipment used in these tests is shown in Figure 10.

Table X lists the results obtained in early experiments. The observed increase in the amount of bismuth distilled when the condenser spacing was increased may be due to an unobserved increase in the surface temperature of the melt due to a reduction in radiant heat loss. Table XI shows data from a run in which the individual cycles were operated at different temperatures. Table XII gives the results of a depletion run with an unrestricted vapor stream while the data in Table XIII were obtained with a 0.063-inch diameter orifice just above the distilling bismuth. The corrected relative volatilities shown in Table XII have been adjusted to account for the amount of depletion found at the end of the run instead of being calculated from the amount of polonium collected.

Table XIV presents the results of an experiment performed to test two-stage simple distillation. These results, compared to previous calculations, 22 indicate that better depletions and product concentrations would be obtained in operation but that greater distilling areas than those calculated would be needed.

Because of the low rate of bismuth vaporization experienced in most of the tests described above compared to that calculated from vapor pressure⁹ data, an investigation was made of factors which may have produced this result. These factors include: (1), contamination of the distilling surface by an oxide film or a film of other impurities originating either in the bismuth charge or from the container material; and (2), a possible difference in temperature between the distilling surface and that measured in the body of the molten bismuth.

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TABLE X

CONSTANT.TEMPERATURE BISMUTH.POLONIUM DIFFERENTIAL.DISTILLATION TESTS

CONC. FACTOR ²									65.5	57.0	45.6	35.8	CONCENTRATION FACTOR
α_1									76.5	70.5	0.06	81.9	RATION
1 B I	0.00184	0 00184	0.00204	0 00207	0.00350	0 00480	0.00526	0 00411	0.00073	0.00080	0.00056	0.00076	CTED
00	1, 25	1 26	1 35	1.37	2.32	3 18	3 23	2 75	0.41	0.62	1.74	2.40	Bi Po NiT. Po OLLE
Bi CHARGE (g)	97.68	96.82	1003	100	100	1003	98.65	98.63	118.93	85.6	86.4	84.43	(2) CF
Po COLLECTED (%)									27.1	35 3	79.2	86.5	
Po CHARGE (C)									0.569	0.410	0.413	0.404	
$TIME \ (h au)$	_	-	-	-	-		-	-	_		4	4	Y1 I.
PRESSURE (µ)	0.026 - 0.05	0.028 - 0.05	0.025 - 0.50	0.045 - 0.40	0.046 0.1	0.042 - 0.42	0.025 - 0.06	0 071 - 0.11	C.	F.O 5.3		5.1	- = RELATIVE VOLATILITY
TEMP (C)	745	745 (745 (745 (745 (745 (745 (745	— = REL
DISTANCE MELT-CONDENSER (in)	_	_	. <u>-</u>		4	4	4	₹ ₹	· -	- -	-	- -	$\alpha = \frac{Po \text{ INIT.}}{B1 \text{ INIT.}}$ $Log B1 \text{ FINAL}$
RUN	-	٠ ،	1 (1	. 4	ى ٠	, (c	7	~ 0	0 0	ກູ	2 ;	12	Ξ

(3) NEW BISMUTH USED IN RUNS 3 THROUGH 6 AND 9 THROUGH 12 INC.

TABLE XI

VARIABLE TEMPERATURE BISMUTH.POLONIUM DIFFERENTIAL-DISTILLATION TESTS

RUN	DISTANCE MELT-CONDENSER (in)	TEMP (0°)	PRESSURE (µ)	Т I МЕ (<i>h</i> т)	Po COLLECTED (C)	Po CUM. (%)	Bi COLLECTED (g)	Bi CUM. (%)	BISMUTH RATE $(1b/in^2/\hbar t)$	a. per cycle	CONC. $lpha$ FACTOR for cycle for oycle
CYCLE 1	-	630	4.3	- ~	0.0875	17.45	0.062	0.057	0.00031	333	306
CYCLE 2	-	029	4.5	101	0.2262	62, 60	0.535	0.550	0.00265	162	Ξ
CYCLE 3	-	710	4.2	- ~	0.1692	96.40	4.8	4.96	0.02380	50.5	20.3
CYCLE 4	<u>-</u>	750	4:2	-12	0.01825	88.66	18.5	22.0	0.09176	18	5.4
CHARGE	108.8 6 8	G B!SMUTH	Į,	0.5017	0.5017 C POLONJUM						

TABLE XII

DATA FOR DIFFERENTIAL DISTILLATION AT 750°C WITH UNRESTRICTED VAPOR STREAM

FACTOR CUM.	37	32	22	0
CONC.	37	63	8	17
CUM. a	74.5	77	75	99
lpha CYC. CUM.	99	99	56.5	8.9 29
CYC.	99	99	8	8.9
RATE Bi DIST. (1bs/in²/hr)	0.0026	0.0005	0.0019	0.0027
.1ST. CUM. (%)	2.14	2.71	4.27	9.7
B1 DIST. CYC. CUM. (\$) (\$)	2.14	0.59	1.6	5.8 9.7
Po DIST. CYC. CUM. (%) (%)	79	87	95.5	6.66
Po CYC.	79	37	84	86
TIME CUM. (hrs)	2	ល	7	12
PRESS. (μ)	25	-	_	-
RUN	4	1 B	10	1 0

TABLE XIII

DATA FOR DIFFERENTIAL DISTILLATION AT 750°C WITH RESTRICTED VAPOR STREAM

			а.	c	m	į		J	7	CONC.	FACTOR
RUN	PRESS (1L)	TIME CUM.	CYC, CUM, (%)	CUM.	CYC, CUM.	CUM.	RATE B1 DIST. $(1bs/in^2/hr)$	CYC.	CYC. CUM.	CYC.	CO
2A	; c o	2	1.3	1.3	90.0	90.0	6.7×10^{-5}	34	34	22	22
2B	15	വ	2.3	3.6	0.016	0.08	1 × 10 ⁻⁵	175	47	142	
*	-	σ	7.2		6.7		5×10^{-2}	18		10.7	
20 20	-	12	0.75		0.04		2.2 x 10 ⁻⁵	19		19	

*THIS PORTION OF THE RUN WAS OPERATED AT SOME TEMPERATURE OVER 900°C BECAUSE OF A THERMOCOUPLE FAILURE

TABLE XIV

TWO-STAGE SIMPLE DIFFERENTIAL DISTILLATION

PRESSURE - 1 TO 10 MICRONS TEMPERATURE - 750°C

	Bi RATE $1b/in^2/\hbar r$) (% of calc)	12.1	9.1	5.0	5.5
	Bi RATE $(1b/in^2/\hbar\tau)$ (%	0.0057	0.0043	0.0024	0.0026
LLED	CYC. CUM. (%)	15.2	1.8	3.32	6.0
Bi DISTILI	cYc. (%)	15.2	1.8	1.55	2.84
o LLED	CUM. (%)	66.66	0.69	86.0	97.3
Po DISTILI	CYC. CUM. (%)	66.66	0.69	54.3 86.0	81.0
CONCENTRATION FACTOR	CUM.	6.6 6.6	38.3 38.3	56	28.5 16.2
CONCENT	cyc.	6.6	38.3	35	28.5
RELATIVE VOLATILITY	CUM.	53	99	57.7	25
RELA	cyc.	53	99	49	46.4
	TIME (ht)		0.16	0.25	0.42
	RUN	44	4B,	, 48,	4B ₃

NOTES - RUN 4A IS FIRST STAGE OF DISTILLATION AND RUNS $4B_1$, $4B_2$, AND $4B_8$ ARE SUCCESSIVE PORTIONS OF THE SECOND-STAGE DISTILLATION.

THIS TWO-STAGE DISTILLATION THEREFORE DIVIDED THE ORIGINAL FEED INTO THREE PORTIONS AS FOLLOWS:

O/Bi) INITIAL	118		
% OF ORIG. PO (Po/Bi) FINAL/(Po/Bi) INITIAL	0.0118	107	0.19
% OF ORIG. Po	0.01	97.3	2.7
% OF ORIG. Bi %	84.8	0.91	14.3
	DEPLETED BISMUTH	PRODUCT	MIXTURE TO BE RECYCLED
	<u></u>	=	=

Table XV lists the rates of bismuth distillation under various conditions of charge preparation and from distilling-cups of different materials. It was concluded: (1), that it is important to use large pieces of bismuth in the charge in order to avoid surface oxidation; (2), that filtering of the charge is not necessary; and (3), that if stainless steel is employed as a cup material it should be pre-heated to clean the surface. While most of the data shown in this table were obtained with inactive bismuth, irradiated bismuth was employed in runs marked 9 and 10. It was found that the relative volatility observed during these runs was the same as that observed at this temperature in previous work where a much lower rate of bismuth distillation was obtained. This indicates that the higher rate of bismuth distillation was accompanied by a similarly increased rate of polonium vaporization.

TABLE XV

TESTS OF RATE OF DISTILLATION OF BISMUTH AT 750°C AND 1 TO 10 MICRONS PRESSURE

RUN	TIME (hr)	CUP MATERIAL	Bi CHARGE	Bi RATE (lb/in²/hr)	PER CENT OF CALCULATED	CONDITION OF CHARGE
1	4.67	STAINLESS STEEL	68	0.0023	4.8	FILTERED, MILLED CHARGE
2	4.25	L. C. STEEL	73	0.0038	8.0	FILTERED, MILLED CHARGE
3	4.0	STAINLESS STEEL	139	0.00037	0.8	REFILTERED, MILLED CHARGE
4	6.0	QUARTZ	56	NEG		REFILTERED, MILLED CHARGE
5	2.0	QUARTZ	30	0.019	41.5	UNFILTERED, LARGE CHUNKS
6	1.75	STA!NLESS STEEL	44	0.01	21.7	UNFILTERED, LARGE CHUNKS
7	2.0	QUARTZ	28	0.02	45.0	FILTERED, LARGE CHUNKS
8	2.0	STAINLESS STEEL	25	0.011	23.6	FILTERED, LARGE CHUNKS
9	1.33	STAINLESS STEEL	84	0.017	37.0	UNFILTERED, PREHEATED CUP
10	0.25	STAINLESS STEEL	39	0.013	27.0	UNFILTERED, PREHEATED CUP

Determinations were made of the temperature difference between the distilling surface and the body of the bismuth melt by calculation and by direct measurement. While not of a high order of precision these results indicated that the temperature of the distilling surface was at most 15°C below that measured in the body of the melt. This difference is not sufficiently large to have been a major factor in producing the low vaporization rates of the previous experiments.

INVESTIGATION OF MATERIALS OF STILL CONSTRUCTION

One of the major problems facing this project has been locating or developing a satisfactory material for construction of bismuth stills. The ideal material for the construction of stills for bismuth-polonium mixtures should have at least the following characteristics:

- (1) It should be capable of containing bismuth up to 850°C or higher under reduced pressure without adding impurities to the bismuth and without being attacked by the bismuth.
- (2) It should be capable of containing polonium under the above conditions without absorbing or combining with small quantities of polonium.
- (3) It should be strong at high temperatures, readily available, and capable of being fabricated easily into vacuum-tight vessels.
 - (4) It should have a low vapor pressure at high temperatures.
 - (5) It should not be oxidized by air at 900°C.

Other desirable characteristics depending upon conditions include immunity to the effect of rapid temperature changes, non-magnetic properties if magnetic agitation of the still contents is employed, and resistance to nitric acid to permit of leaching periodically.

CORBOSION BY LIQUID BISMUTH

The corrosion of metals and other materials by molten bismuth has been investigated fairly extensively by others. A full treatment of the work at other sites is beyond the scope of this report and the reader is referred to Reference 16 for a review of the subject. Additional information has been published 17,18 recently.

At our request Battelle Memorial Institute tested a number of materials for corrosion by bismuth under specified conditions in order to make the data applicable to the distillation project. Their work 19 covered static corrosion of about 50 materials by liquid bismuth at 750°C and 850°C. These materials included low-alloy steels, silicon steels, stainless steels and super heat-resistant alloys. The metals were examined visually and metallographically after exposure to the bismuth and the bismuth was analyzed for impurities absorbed from the alloy. In fairly long-term tests Armco iron, low-carbon steels and chromium-plated samples were found to be severely attacked while the chromium alloys, stainless steels and superalloy steels were moderately attacked. A 2.9 per cent silicon transformer steel and Duriron were the only materials tested which exhibited good corrosion-resistance to bismuth at 750°C and 850°C.

At Mound Laboratory corrosion tests of bismuth on various materials have been made over a period of time. Many of the experiments listed below were performed before portions of the reference material described above became available.

R. W. Moshier of the Chemistry Section reported some early corrosion experiments. 20, 21 Table XVI lists these data for the corrosion of low-carbon steel by bismuth. The first eight tests were performed by refluxing bismuth in a steel cylinder in a residual nitrogen atmosphere of about one millimeter of mercury pressure. A "condensation ring" containing a brittle deposit was found in all cases on the tube wall just above the heated zone. Although the deposit appeared to be an alloy, bismuth flowed out upon heating a portion of it to 1000°C, leaving a residue of the same form as the original piece. This residue is thought to be a spongy deposit of condensed iron containing absorbed bismuth.

TABLE XVI

CORROSION OF LOW-CARBON STEEL BY MOLTEN BISMUTH

							IRON IN
TEST NO.	BISMUTH CHARGED (g)	TEMPERATURE (°C)	VACUUM (mm Hg)	DURATION OF TEST $(h au)$	BISMUTH RECOVERED (g)	IRON IN BISMUTH (%)	CONDENSATION RING (%)
1	100	980	1	24	81	0.005	•
2	100	980	1	24	93	0.005	28
3	100	850	1	48	73.7	0.006	•
4	100	850	1	48	78.0	0.003	15.5
5	100	850	1	78	78.0	0.05	<u>-</u>
6	100	850	1	78	TUBE FAILED WHE	N WELD OPENE	30
7	100	850	0.7	100	80.0	0.2	•
8	100	850	0.7	100	85.8	0.24	26**
9*	100	850	0.7	100	97	0.08	0.04

^{*}IN QUARTZ TUBE WITH PIECE OF STEEL OF AREA OF 3.34 SQUARE CENTIMETERS.

Table XVII contains the results of tests on miscellaneous metals made by exposing the sample to bismuth contained in a quartz tube. This work was the first indication that silicon had a beneficial effect on the corrosion resistance of iron alloys by bismuth. Table XVIII lists data obtained on the corrosion-resistance of high-temperature alloys. These tests indicate that alloys S-590 and 19-9DL have fairly good resistance to corrosion.

Tests were performed by the Chemical Engineering Section^{5, 22} in order to specify materials for experimental still construction. Two tests on SAE 1020 steel tubes containing bismuth for 100 hours at 900°C and one micron pressure gave 0.3 per cent and 0.1 per cent of iron in the discharged bismuth. ²² A similar test on a high-chromium steel (18% Cr. 0.65% Mo) resulted in bismuth containing 2.4 per cent of chromium.

^{**}WE!GHT OF CONDENSATION RING. 17.5 GRAMS.

TABLE XVII

CORROSION TESTS ON MISCELLANEOUS METALS*

TEST METAL	TEMPERATURE (°C)	VACUUM (microns)	TIME (<i>ħτ</i>)	RESULTS
TANTALUM	850	50	100	THE 1-INCH SQUARE SHEET OF 6-MIL Ta LOST 51.5 MG. THE IMMERSED PORTION WAS VERY BRITTLE.
AL i NCO	850	50	24	DURING THE TEST NEEDLES APPEARED ON THE BISMUTH SURFACE AND SOLIDIFIED. A 35° INCREASE IN TEMPERATURE PERMITTED CONTINUATION. NI, Fe, Cu, Co were present in the BISMUTH.
MOLYBDENUM	900	5υ	190	ALL THE BISMUTH HAD CONDENSED IN THE COOL PART OF THE CHAMBER AT THE END OF THE TEST. A SPECTROGRAPHIC TRACE OF MOLYBDENUM WAS PRESENT IN THE BISMUTH. THE IMMERSED PART OF THE MOLYBDENUM WIRE WAS BRITTLE.
DURIRON (TEST 1)	850	50	72	0.002 PER CENT IRON WAS FOUND IN THE BISMUTH POOL, AND 0.001 MG IN THE BISMUTH MIRROR ON THE WALLS OF THE CONTAINER.
(TEST 2)	850	50	100	A TRACE OF IRON AND OF MOLYBDENUM WAS PRESENT IN THE BISMUTH. THE 25 G TEST PIECE LOST 21.5 MG.

^{*100} G OF BISMUTH WAS CHARGED IN EACH TEST.

A series of short-time screening tests on 14 alloys were performed in quartz tubes with bismuth heated by induction heaters. The alloy composition, final bismuth composition and the sample weight-loss are listed in Table XIX. The three samples considered worthy of further investigation were the super-alloys 19-9DL, S-590 and Haynes Stellite 25. The results of further checks on these materials are given in Table XVIII.

TABLE XV!!!

CORROSION TESTS ON HIGH-TEMPERATURE ALLOYS

	TEMP (°C)	;	880	875	890
AP E AR ANCE	AFTER TEST		ETCHED	CORNERS ALLOYED	NO CHANGE
CORROSION RATE	(g/in ² /100 hr)		0.0678	(0.907 IN2 AREA)	0.0532
	CHANGE (g)		0.1560 Loss	0.0022 GAIN	0.0630 Loss
WEIGHT	AFTER (g)		8.6225	9.3947	8.7370
	BEFORE (g)		8.7785	9:3925	8.8000
ALLOY DESIGNATION		HAYNES STELLITE	NO. 25	S- 590	19-9-DL

ANALYSIS OF THE BISMUTH

Į į	Ē				NONE
윤 (R		TRACE		TRACE
88	₩ <u></u>		TRACE		
Al	<u>&</u>		TRACE		
Ag	<u>k</u>		TRACE		
g (<u>&</u>		NONE		NONE
≥	&		NONE		NONE
Mo	<u> </u>		TRACE		NONE
ភ	<u>&</u>	0.04	TRACE	6.0.0	NONE
W	&		TRACE		NONE
ŝ			TRACE		WEAK 0.007
బ్ర	(%) NOT TAKEN	90.0	WEAK		
	(%) NOT TAKEN		WEAK	0.016	TRACE
	(%) NOT TAKEN		WEAK		TRACE
HAYNES STELLI'TE	NO. 25 SPECTRO	СНБМ	S-590 SPECTRO	CHEM 19.9.DL	SPECTRO

CHARGE - 100 G BISMUTH

PRESSURE . 124

TABLE XIX

MATERIAL OF CONSTRUCTION TESTS*

	SAMPLE WT LOSS (8)	0.0026 (GA EN)	0 5830	0.0174	0 0037	0.3631		0.1473	0,6072	ī	0.1956	0.0145	0.2317	0.1294	0.2020
	TEMP.	096	1 000	840	900	980	950	920	096	920	925	900	950	950	920
	T.ME	30	35	35	45	37	09	90	90	90	90	30	09	9	9
	PRES. (μ) (-	0.	0.7	5.	 V		·	-	-	<u>-</u>	- V	· -	-	-
	_	<u>-</u>	0	9	-			<u>-</u> ۸							
	SAMPLE WT. (g.)	9.8431	6.9670	9.7126	6 . 3391	4.0553	9.6620	12 2201	12.0450	5,2381	12.6036	7.0940	17.9412	12.9649	14.3350
F Z	ъ. Ф	0	0.04	0	TRACE	0 %	0.33	4.4	7.5	0.70	0.10	0.08	0.04	0.04	0.08
R CE) N	1	1	-	0	0	0	1	1	0	1	1	1	1	i
, H	ಬ	1	0.06	0.0	i	0 05	1	1	ì	1	ı	1	1	1	I
COMPOSITION, PER CENT	ž.	0	0.16	TRACE 0.001	0	0	0.135	0.014	0 077	0.074	90.0	ı	0.02	0.02	0.03
COMP	ភូ	0	0.045	0	0	0.054	0.01	0.02	60 0	0.13	0.009	0.001	0.009	0.016	0.007 0.03
11. B1	87	1	F	0	0	0	9	ī	1	1	ı	1	1	1	1
FINAL	M	1	0.02	1'RACE	0	110	0.063	0.05	0.04	0.05	ı	1	1	ı	1
	ಬ	1	1	1	;	0	!	0	0	0	ı	1	1	1	1
	Б. Ф)	9	REM	5- REM	R E	0.70	4.7	REM	REM.	REM.	REM.	REM.	REM.	REM.	R EM
ED)	*	*	6.4 5.	5.4 5.5	- 2	;	ì	1	1	1	I	I	1	ŀ	!
L 3 SH	I	ı	1	1	0.2	1 -	1	1	1	1	1	1	0.4	ŀ	ı
CENT (PUBLISHED)	ឡ	1	6. 4 5.	3.5	0.3	i	;	1	0.8	1	ı	1	,	ı	1
R CENT	N.	1	ພ. <u>⊿</u> ຕຸ	ಲ್ಲ.∡4 ಸ್ಥ	1,25	5.5	28.6	ı	1	3.5	ı	ı	ı	1	ı
ON, PER	೪	: ,	42.	19. 21	;	62.2	i	1	ı	ı	ı	1	i	t	1
T:0N	N 1	*	21.	21.	6	8.8	65	. .	9.	<u>0</u> 4	20	1	÷=	19- 22	12.
COMPOSIT:	Ç	:	18.5- 20,5	18.5	6	27.4	1	18. 20	17.	-91	25	23. 27	17.	2 4 . 26	22.
SAMPLE	81	1	0.	0.	0.60	0.53	0.19	ì	1	ı	ı	ı	1	i	1
SAM	=	ŧ	0.	2.0	0.5	0.66 0.53	0.59	2.0	1	ı	1	ı	ı	2.0	2.0
	D .	*	0.35	0.4-	0.25	0.22	0.05	0.08	0.08	0.10	0.10	0.35	0.08	0.25	0.20
	SAMPLE ASM CODE	HAYNES STELL!TE 25	HAYNES STELL: TE S-816	SUPER ALLOY S.590	SUPER ALLOY 19-9 DL	V:TALL!UM	HASTALLOY B	STA!NLESS STEEL-304	STAINLESS STEEL-347	STAINLESS STEEL-316	STAINLESS STEEL-314	STA!NLESS STEEL-446	STAINLESS STEEL-321	STAINLESS STEEL-310	STAINLESS STEEL-309

^{*}APPROXIMATELY 100 GRAMS OF BISMUTH WAS USED IN EACH TEST

^{**} AMOUNT UNKNOWN

COMBINATION OF POLONIUM WITH MATERIALS OF STILL CONSTRUCTION

Very little information is available on the combination of polonium with materials of still construction Investigations²³ by the Physics Section indicate that polonium forms compounds with zinc lead platinum nickel silver and sodium but no compounds with gold, tantalum, iron beryllium and aluminum were found A test²² performed to simulate still conditions consisted of sealing one curie of polonium in a small steel container and repeatedly heating and cooling it. It was opened and heated to a high temperature to drive off the free polonium. Upon dissolving the container 2.6 per cent of the original polonium was found. A second test consisted of condensing on and driving off polonium from a steel sample until a total of 15 curies had been so transferred. Upon calorimetering the steel sample it was found to retain about one millicurie. It was concluded that retention of polonium by steel was low.

CREEP-STRENGTH AND PROTECTION AGAINST SCALING

Low-carbon steel was used as a material of construction for experimental stills until a better material could be found, although when used alone it has major disadvantages. It has low creep-strength at elevated temperatures; is subject to rapid atmospheric corrosion at elevated temperatures, and is attacked by bismuth to a certain extent. The disadvantage of low creep strength tends to limit the size of still-pots so that still wall-thickness increases rapidly for stills over about eight inches inside diameter. Atmospheric corrosion on the outside of the still has been a problem. Heavy nickel-plate if properly applied offers protection from scaling of mild steel for several hundred hours at operating temperatures. Aluminum impregnation appears to offer similar protection; however, it may create problems in heat transfer as this type of surface seems to act as a refractory material. Chromium and silicon impregnation appear to be unsatisfactory and both are difficult and expensive operations Silicon-impregnated surfaces are very brittle and are damaged by thermal shock. A protective surface which promises to overcome some of the above disadvantages is that of surfacing by-welding Test pieces of low-carbon steel enveloped in $\frac{3}{6}$ -inch of type 310 stabilized stainless steel weld-stock have stood up well to thermal shock tests. In addition 310 stainless steel has a creep strength of 2300 pounds per square inch for 1 per cent creep in 10 000 hours at 1400°F as compared to approximately 100 pounds per square inch for low-carbon steel. Both literature²⁴ reports and our tests on surfaced samples indicate that type 310 stainless steel is resistant to scaling at 2100°F. The surfacing-by-welding process is expensive and time-consuming for large pieces

GRAPHITE FOR STILL CONSTBUCTION

Graphite¹⁸ is another material which has not been tried for still construction here, but which warrants consideration. The use of a non-metal would involve major revisions in still design but in view of the fact that no ideal metal has been found, it may provide the best long-range solution to this problem.

CURRENT PRACTICE

Although perhaps not an ideal system, low-carbon steel with a welded type 310-Cb stainless steel covering will be used for the pilot plant still because the low-carbon steel has stood up fairly well as the interior of experimental stills and it does not appear to absorb polonium. The welded stainless steel cover provides high-temperature strength and oxidation resistance.

STILL DESIGN AND CONSTRUCTION PROBLEMS

SEALS

Faulty vacuum-tight seals were troublesome in early still designs. Attempts to use metal seals near heated zones were generally unsuccessful. However, by allowing air-cooled or water-cooled sections to isolate the seal from a hot zone. "O" ring seals proved to be fairly reliable. Since bismuth expands upon solidification, this property has been used in the sealing of bismuth flow-lines. A line sealed in this fashion usually gives an excellent vacuum seal which can be opened readily by application of heat. "Weather-head" compression fittings have been successful for some sealing applications but were not satisfactory for sizes larger than ½ inch. A modification of this fitting is the "Swagelok" compression fitting. This type of fitting appears to give a more positive seal at elevated temperatures and has been used with success at Brookhaven National Laboratory for vacuum-tight tubing connections.

PRODUCT CONDENSERS

Product condensation and collection is a continuing design problem. Solid bismuth is a notoriously poor conductor of heat Also, the bismuth vapor condenses as a dendritic needle-like structure which further reduces the heat-transfer characteristics of the product mass. These conditions give rise to three types of troubles: (1), the product crumbles fairly easily if sharply jarred (2) product build-up may be limited by its poor heat-transfer property; and (3) bridging of the product to the column side-wall must be avoided in condenser design and operation

Current condenser design provides for a cooled cylinder suspended in the heated upper portion of the column in such a fashion that an extension of the column extends a short distance into the condenser. The expanded portion of the column into which the condenser fits and the column extension provide an annulus into which the lower portion of the condenser is suspended without making contact with the hot column side-walls as shown in Figure 5. This arrangement has permitted separation of the product from liberated or introduced gases and collection of the product within the condenser which results in material balances being well within experimental counting error. However, bismuth which drops into this annular space due to spattering or thermal short-circuiting of product is difficult to remove, and if left, causes low concentration-factor results for the products of succeeding distillation runs. Condensers of this type also introduce undesirable restrictions in the vacuum system. Consequently a study of methods of product collection should continue to be made until these conditions are corrected.

Another condenser design which is being considered would have the vacuum drawn directly through the cooled collector past baffles. There would be less chance with such a design of bismuth or polonium accumulating in the annular space.

SPARGERS

Sparger design has proven troublesome. If incorporated into the still as a permanent non-replaceable part the sparger becomes the most vulnerable spot in the still design. If incorporated as a replaceable part, problems of gas dispersion and vacuum-tight seals arise.

A sintered disk in the bottom of a still proved unreliable due to fabrication difficulties. However, a test piece with a sintered disk made to observe gas distribution within the bismuth melt showed excellent distribution. It permitted helium to pass into the melt while at the same time acting as an effective barrier with or without gas flow to the reverse flow of the molten bismuth.

A perforated disk with perforations of 0.025-inch diameter was effective. After many runs with this type of disk most of the perforations were clogged yet no difference was noted in distillation rates or product concentration.

WELDED JOINTS

Most still failures can be attributed to welds which for one reason or another became porous after the still was put in operation. Familiarity with good welding technique is essential for the designer as well as for the welder. Common mistakes in design include lack of provision for welding room between parts; welding of unequal masses; intricate joints which cause non-continuous welding and excessive metal stressing; and the use of welded joints where a completely machined part could be used or where parts could be joined effectively by silver solder or by mechanical seals. Common mistakes in welding include improper choice of welding rod; lack of proper equipment; shallow welds; scale and flux inclusions; thermal shock; and no provision for annealing after welding. Vacuum—tight welded joints for low-carbon steel can be obtained by: (1) using either Heliarc or Sigma welding equipment; (2), providing an inert atmosphere in the interior of cylindrical parts during welding; (3), providing for 100 per cent welds instead of shallow welds: (4), using welding rod of the same composition as the stock to be welded; and (5), making provision for a slow cooling cycle if the part is not immediately annealed after welding.

SUMMARY

Early bismuth-polonium stills indicated the possibility that distillation could be developed into a practical process for the separation of polonium from bismuth. However, the first glassware-contained experiments and the first fractionating still gave fair and excellent concentration factors, respectively, at low depletion rates. The stripping still, which was operated at a higher temperature, depleted the bismuth rapidly at the expense of product concentration.

Agitation by sparging was investigated as a means of improving still performance and appeared to produce considerable benefit. Two stills employing this principle and differing only in their material of construction were built and produced excellent results. The bismuth charge was usually depleted of over 99.5 per cent of its polonium in three hours and a product concentration 150 to 1000 times that of the original charge was obtained.

Simple distillation without fractionation was investigated. Considerable work was done on the relative volatilities and vaporization rates of polonium and bismuth from their mixtures. The results obtained with small experimental units indicated that a two-stage system of simple stills of reasonable size operating on a one-day cycle could give a concentration factor of several hundred while recovering well over 99 per cent of the polonium from irradiated bismuth.

Finding a suitable material for the construction of stills has been a major problem facing this project. Resistance to molten bismuth, no affinity for polonium, and high temperature strength are some of the necessary properties. No completely satisfactory material has been located either by this Laboratory or by other sites which have investigated this problem. At our request Battelle Memorial Institute did considerable work on this phase of the project and recommended a silicon steel as having the best resistance to bismuth of the many alloys

that they tested One of the sparging stills previously mentioned was constructed of this material but gave indication of absorbing an excessive amount of polonium on the silicon steel walls. Currently stills are being constructed with a low-carbon steel body with an outer layer of stainless steel applied by welding to give strength and resistance to high temperature scaling

In the process of constructing and operating experimental stills considerable information and experience has been gained on the various phases of still design. High temperature seals product condensers sparging inlets and welded joints have all presented special problems which have been solved to a reasonable degree

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